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Mat. Res. Bull. Vol. 6, pp. 433-440, 1971. Pergamon Press, Inc. Printed in the United States.

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HIGH PRESSURE SYNTHESIS AND CRYSTAL DATA OF THE RARE EARTH ORTHOALUMINATES

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(Received March 25, 1971; Refereed)

ABSTRACT

Single crystals of the rare earth orthoaluminates, REAlO₃, with RE = Dy - Lu, have been synthesized at high pressure in the presence of a flux. Lattice parameters for the orthorhombic members of the series, Sm-Lu, are reported. The a and c lattice parameters decrease smoothly in the direction Sm to Lu but the b parameter increases asymptotically to a maximum value at Lu. The anomalous behavior of the b parameter is similar to the variation of this parameter in the isostructural REFeO₃ series between La and Gd. For these latter compounds the variation has been explained in terms of a gradual change in the coordination number of the rare earth cation. A similar mechanism appears to be the cause in the case of the rare earth orthoaluminates.

Introduction

Unlike the ABO₃ perovskite-like compounds where A = rare earth and B = Fe, Ga, V, Cr, Rh, Sc and In, the rare earth orthoaluminates are not isostructural across the entire series. From La through Nd the orthoaluminates have a trigonal distortion of the ideal perovskite structure whereas from Sm through Lu they have the orthorhombic distortion found in the rare earth orthoferrites. The series is also unusual in that the synthesis of the members Dy-TmAlO₃ at atmospheric pressure generally yields a mixture of phases, namely the rare earth orthoaluminate plus the rare earth aluminum garnet(1). More

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gallium garnets and Ga203(7), and for this reason was chosen as a flux for the orthoaluminates. Sample volumes of approximately 0.054 cc. each were packed into separate platinum containers. During all runs, two materials were pressed simultaneously at 32.5 kilobars and 1200°±10°C, for a period of four hours. After quenching, the apparatus was allowed to cool over night, the pressure released, and the subsequent recovery of the sample containers effected. Each platinum cell was then broken open and immersed in distilled water. The NaOH rapidly dissolved leaving small, rectangular parallelpiped-shaped, transparent crystals. The crystals of the Ho, Er, and Tm compounds were colored pale yellow, pale pink, and pale green respectively, while those of Dy, Yb and Lu aluminate were colorless. All the crystals were strongly birefringent.

X-ray powder films were taken of Eu-LuAlO3 crushed crystals with a Guinier camera using KCl as internal standard and CuKa radiation (Ka₁ = 1.5405 A and Ka₂ = 1.5443 A). None of the powder films showed any trace of garnet lines after 24 hrs. exposure. They were all indexed on the basis of the orthorhombic cell, space group Pbnm, with lattice constants given in Table 1.

TABLE 1

Orthorhombic Cell Data for the Series SmAlO3-LuAlO3. The standard deviations are less than 10-4, except for those of EuAlO3 and GdAlO3 which are twice as large.

	0	0	0	0
	<u>a,(A)</u>	b,(A)	c,(A)	Vol. A
SmAlO3 EuAlO3 GdAlO3 TbAlO3 DyAlO3 HoAlO3 ErAlO3 TmAlO3 YbAlO3 LuAlO3	5.2912 5.267 5.250 5.2317 5.2053 5.1811 5.1595 5.1435 5.1251 5.1012	5.2904 5.294 5.302 5.3097 5.3172 5.3229 5.3271 5.3277 5.3210 5.3310 5.3317	7.4740 7.459 7.447 7.4196 7.3950 7.3741 7.3539 7.3335 7.3146 7.3000	209.22 208.0 207.3 206.11 204.68 203.37 202.12 200.96 199.85 198.55

The lattice parameters were refined from values of 2θ which were $\leq 2\theta(224)$, excluding overlapped lines. The lattice parameters for SmAlO3 were determined by a simplified version

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of Bond's method (8,9). The powder patterns for Eu-LuAlO3 are listed in Table 2.

Discussion

The lattice parameters of $SmAlO_3$ -LuAlO₃ are plotted against atomic number in Fig. 1. Atomic numbers rather than ionic radii were chosen as abscissae since the variation in coordination number across the series is not known. One can see that while the <u>a</u> and <u>c</u> parameters decrease in a nearly linear fashion from Sm-Lu, <u>b</u> increases asymptotically to the Lu value. The behavior of the <u>b</u> parameter resembles that of REGaO₃(10)compounds between Ce and Gd and the <u>b</u> parameter of the REFeO₃ series between La and Gd(11). However, in the case of the two latter series, the <u>b</u> values eventually decrease for smaller rare earth ions, but no decrease occurs for the REAIO₃ compounds.

The entire series of REFeO₃ structures have been refined from single crystal data in order to fully understand the mechanism underlying this anomalous behavior(4). It is now clear that the oxygen polyhedron around the rare earth ion is distorted and this distortion varies drastically across the series. The rare earth ion has approximately an eight fold





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TABLE 2

				m-		D		Ho		P				v		Te	Ta)
	4						2		4	E	4			4	4		
0	ac	°o	de	ao	ac	do	dc	do	de	o	de	u o	de	^d o	de	do	°°.
								0. 711		2 000	2 70(4.215 vw	4.211	4.203 VW	4.197	4.180 VW	4.131
735 m	3.734	3.731 m	3.731	3.729 m	3.727	3.722 m	3.720	3.714 B	3.713	3.709 m	3.100	3.700 ms	3.100	3.090 5	3.095	3.005 ms	3.650
	3.131	3.720 1	3 - 123	3.111 Wm	3.110	3.099 1	3.090	3.000 ms	3.00/	3.019 m	3 310	3 303 1	3 304	3 208 m	3.208	3.291 m	3.200
647 H	2.647	2.651 1	2 651	2 653	2.655	2 650 Mm	2.650	2 662 100	2.661	2 663 100	2.664	2.663 m	2.664	2.665 m	2.666	2.665 vm	2.666
	2 630	2 635 4	2 635	2 628 0	2 620	2 622 VG	2 622	2 617 ve	2.616	2.611 6	2.610	2.603 8	2.605	2.598 45	2.599	2.593 VS	2.594
.6 3 9 s	2.634	2.626 4	2.625	2.616 4	2.616	2.603 4	2.603	2.501 m	2.591	2.579 Wm	2.580	2.572 m	2.572	2.562 m	2.563	2.550 Wm	2.551
	2.034	2.400 100	2.407	2.500 1	2.500	2.502 Wm	2.502	2.503 Vm	2.503	2.505 m	2.504	2.503 m	2.504	2.503 m	2.504	2.503 m	2.504
			21491	21,00 10	21,000	L.JOL WI	2.702	2.705 ##	21/05	2., joj m	2.214	2.208 W	2.209	2.202 W	2.202	2.194 W	2.194
										2.214 w	2.214						
158 mm	2.159	2.159 W	2.159	2.159 W	2.159	2.158 m	2.159	2.159 m	2.158	2.158 wm	2.157	2.154 m	2.155	2.154 m	2.154	2.153 m	2.153
151 wm	2.151	2.145 W	2.145	2.138 w	2.138	2.128 m	2.128	2.120 m	2.120	2.113 wm	2.112	2.104 m	2.105	2.098 ms	2.099	2.091 m	2.091
				2.060 VW	2.061	2.055 VW	2.055	2.050 w	2.050	2.044 w	2.045	2.039 wm	2.040	2.035 m	2.035	2.030 m	2.031
						and the second		1.992 VW	1.992					1.986 vw	1.986		
965	1.867	1.865 wm	1.865	1.864 wm	1.863	1.860 m	1.860	1.856 m	1.856	1.854 wm	1.853	1.850 m	1.850	1.848 m	1.847	1.843 m	1.843
005 WE	1.865			1.855 w	1.855	1.849 m	1.849	1.843 m	1.844	1.839 wm	1.838	1.833 wm	1.833	1.828 m	1.829	1.824 wm	1.825
815	1.812			1 808	1.810	1 805	1.807	1.805 W	1.806	1.803 vw	1.804	1.802 wm	1.801	1.799 wm	1.799	1.797 wm	1.797
01) W	1.811			T*000 M	1.807	1.005 Wm	1.804	1.800 wm	1.800	1.797 m	1.797	1.795 wm	1.794	1.791 m	1.791	1.787 m	1.787
668	1.669	1 665	1.668	1.665 VW	1.665	1.661 vw	1.661	1.658 w	1.658	1.654 wm	1.655	1.652 w	1.652	1.649 W	1.649	1.645 wm	1.645
000 W	1.668	1.003 WI	1.666	1.661 vw	1.661	1.655 w	1.656	1.652 wm	1.651	1.647 wm	1.647	1.643 wm	1.643	1.639 wm	1.639	1.635 m	1.636
				1.657 VW	1.657	1.649 vw	1.649	1.643 vw	1.643	1 637 m	1 637	1 636 m	1.636	1.628 w	1.627	1.620 w	1.620
		1.634 w	1.634	1.635 vw	1.635	1.636 wm	1.636	1.636 wm	1.637	7:031 m	1.031	1.030 11	1.632	1.636 m	1.636	1.635 m	1.636
											10.000	1.593 w	1.593	1.589 vw	1.588		
		1.527 w	1.527	1.528 w	1.528	1.528 wm	1.528	1.527 wm	1.528	1.527 wm	1.527	1.527 wm	1.526	1.526 wm	1.526	1.526 w	1.525
				1.520 w	1.521	1.518 wm	1.518	1.515 wm	1.515			1.510 wm	1.510	1.508 wm	1.508	1.506 wm	1.506
522 W	1.522	1.518 wm	1.519	1.512 m	1.513	1.507 m	1.507	1.502 w	1.502	1.497 wm	1.497	1.493 w	1.493	1.489 w	1.489	1.485 wm	1.484
	1.521		1.518		1.513		1.506	1.500 m	1.501	1.495 m	1.495	1.491 m	1.491	1.487 m	1.486	1.482 wm	1.481
						0	0.0		2 201	1.478 w	1.478	1.476 wm	1.475	1.473 wm	1.472	1.470 w	1.469
					1.387 w	1.387	1.386 wm	1.386	1.384 m	1.385	1.384 m	1.384	1.383 wm	1.383	1.382 wm	1.382	
													1.360 vw	1.360			
								2 200		1.310 W	1.310	1.310 VW	1.310	1.312 W	1.311		and the second
320 W	1.319			1.315 W	1.315	1.311 m	1.311	1.308 m	1.308	1.305 m	1.305	1.302 wm	1.302	1.300 wm	1.300	1.297 m	1.297
								1 000	1 003					1.215 wm	1.215		
						1 178 -	1 176	1.220 W	1.221				1 170	1 169	1 167		
						1.10 m	1.110					1.169 wm	1.170	1.100 Wm	1.107		
												1 167	1.109				
	1 180											1.10/ W	1.107	1 159	9.350		
179 w	1.100											T*100 MW	1.101	1.130 W	1.150		
	1.119											1 140	1 140	1 118	1 11.7		
												1.149 W	1 104	T. 140 A	1.14/		
												1.105 wm	1.104				
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coordination between TbFeO₃ and NdFeO₃. At the extremities of the series this approximation breaks down. For LuFeO₃ the seventh and eighth nearest oxygen atoms are becoming second nearest neighbors, while for LaFeO₃ the ninth nearest oxygen is too close to be considered a next nearest-neighbor. This change in coordination number governs the behavior of the <u>b</u> parameter.

It seems likely that a similar mechanism applies in the case of the REALO₃ series. However, it is important to note that the orthorhombic series begins with SmALO₃ where the coordination number of Sm³⁺ is very nearly twelve, compared to eight for its iron counterpart. Also, the nonlinear variation of the <u>c</u> parameter and the significant change in slope of the b parameter between Sm and Tb in Fig. 1 suggest a rapid decrease in the coordination numbers of the rare earth ions. Between DyALO₃ and LuALO₃ the coordination number does not appear to decrease as drastically. However, without a detailed knowledge of the structure of at least several more REALO₃ members, it is difficult to ascertain how the rare earth polyhedron varies across the series.

Another interesting point is that starting with Ho one needs high pressures to synthesize single phase rare earth orthoaluminates. $LuAlO_3$ was easily formed at 32 kbar but no attempt was made to find the minimum pressure necessary for this synthesis. We suspect that 32 kbar exceeds the minimum considerably. It is a logical step to attempt to synthesize under pressure MALO₃, where M is of smaller ionic radius than Lu^{3+} . We believe In³⁺ and possibly Sc³⁺ are likely M-cations and expect to proceed with these experiments in the near future.

Acknowledgements

The authors wish to thank J. P. Remeika for the Sm-TbAlO₃ crystals used in the lattice parameter determinations and M. Marezio for helpful discussions and continued interest in this work.

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